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U. S. NAVAL PROVING GROUND

DAHLGREN, VIRGINIA

NPG REPORT NO. 3-47

METALLURGICAL EXAMINATION OF STANDARD U. S. NAVY ARMOR-PIERCING PROJECTILES

8#	AP	PROJECTILE	MARK	21-5
		PROJECTILE		
		PROJECTILE		
16"	AP	PROJECTILE	MARK	8-6

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#### METALLURGICAL EXAMINATION OF STANDARD U. S. NAVY ARMOR-PIERCING PROJECTILES

8" AP PROJECTILE MARK 21-5 12" AP PROJECTILE MARK 18-1 14" AP PROJECTILE MARK 16-8 16" AP PROJECTILE MARK 8-6

1. The results presented in this report are a part of the general program of development of armorpiercing projectiles being carried out by the Armor and Projectile Laboratory at the Naval Proving Ground.

2. The projectiles were sectioned and the surfaces ground at the Naval Gun Factory. The metallurgical examination was performed by the author and other staff members of the Armor and Projectile Laboratory.

JOY

Rear Admiral, USN Commanding Officer

## PREFACE

#### AUTHORIZATION

Specific directives for this investigation were issued in BuOrd ltr. NP7/S78 (Re3) of 29 November 1945.

#### OBJECT

This investigation was conducted to obtain metallurgical information on the present standard 8", 12", 14", and 16" AP projectiles.

#### SUMMARY

The examination of the projectiles included the determination of the hardness distribution, the Charpy V-notch properties, the macro- and mirco-structures, and the chemical composition. Results are given and discussed. The projectiles are in general quite similar.

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I.

#### INTRODUCTION

It is usual to subject one representative from each group of experimental projectiles to a metallurgical examination, which includes the determination of the hardness distribution, the Charpy V-notch properties, the macro- and micro-structures, and the chemical analysis. In the past, such examinations have been limited to projectiles smaller than 8", but the acquisition of new hardness testing equipment has made possible the application to projectiles of any size in use. The present standard 8", 12", 14", and 16" AP projectiles have therefore been subjected to examination, in order to provide a comparison for future work with experimental projectiles.

A brief discussion of the significance of the various tests follows.

#### 1. Hardness and Hardness Distribution

The hardness of a projectile cap and also of the nose of the projectile influences its ability to cause damage to the face of Class A armor. The extreme hardness of the cap is instrumental in starting damage to the surface. After the cap has been shattered the nose of the projectile is able to complete the damage started by the cap. The hardness alone is not a measure of the quality of the projectile. The high hardness necessary to cause the plate to fail must be backed up by a more ductile material capable of absorbing shock without failure. The hardness pattern or hardness distribution gives a measure of the contour of the hardened zone.

Work done at the Naval Proving Ground on AP projectiles equipped with caps of hardness greater than 600 Brinell (carbide ball) led to the development by the projectile manufacturers of 6" and 8" AP projectiles with considerably improved penetrative performance against Class A plate over standard projectiles. As a result of this work 6" and 8" AP projectiles are now equipped with caps of approximately 680 Brinell (carbide ball) as contrasted with the old production projectiles which had a cap hardness of 550 (carbide ball) Brinell.

Deeper hardness patterns and higher nose hardnesses for the bodies of AP and Common projectiles have been advocated by the Proving Ground for several years. It appears that the nose hardness should be as high as

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possible while the body hardness should be that which gives the maximum bend strength.

2. <u>Impact Test</u>

The Charpy V-notch impact test is a measure of the ability of the material to absorb shock without fracture. The test is usually applied to a sample from the body of the projectile rather than from the cap or nose. The principal property required of the cap and nose is hardness while the body of the projectile must withstand stresses without failure in order to carry the projectile through the plate.

#### 3. <u>Macroetch</u>

It is well-known that by macroetching important structural features of the material are developed. Any dendritic segregation which may have been retained from the original ingot will be shown. The fiber of the metal which indicates the method of forging will be shown, though this may not be clear in a fine grained well hardened alloy steel. An added feature of the macroetch is its ability to develop the pattern of the hardened zone. Hardened steels show a darker color than unhardened steels when etched with ammonium persulfate.

#### 4. <u>Microstructure</u>

The microstructure shows the general quality of the steel and also the effect of heat treatment. Physical properties may be predicted with reasonable accuracy by studying the structure revealed under the microscope. Unetched specimens are usually used in order to determine the number and size of inclusions. This latter feature is usually of minor importance in well made alloy steels.

#### 5. <u>Chemical Analysis</u>

By chemical analysis the type of material is determined. This information gives a reasonable indication of hardenability and indication of probable physical properties. Certain features such as susceptibility to temper brittleness and possibility of cracking on quenching are also predictable. By determining the amount and kind of alloying elements present it is possible to predict the results obtainable by tempering at various temperatures.

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By examining a projectile with all of the above methods it is usually possible to determine the effectiveness of heat treatment and also to discover if the proper heat treatment has been applied.

### II. <u>DESIGN FEATURES</u>

Outline drawings of the projectiles are shown in Figure 1 and a tabulation of some of their major design features is given in Table I.

#### TABLE I

8" AP 12" AP 14" AP 16" AP Mk, 21-5 Mk. 18-1 <u>Mk. 16-8</u> Mk. 8-6 8 16 Diameter (in.) 12 14 Overall length 36 54 56 72 (in)49.8 Body length (in) 23.2 37.7 36.5 6.8 5.5 6.0 Cap length (in) 5.1 Loaded wt. (lbs.) 1140 1500 2700 335 Body wt. (lbs.) 245 1203 2191 925 Cap wt. (lbs.) 58 124 153 312 % Cap by wt. 17.3 10.9 10.2 11.6 % Bursting charge 1.5 1.5 1.5 1.5 by wt. 578 578 Maximum cap hard-683 555 ness (BHN) 1.2 2.3 1.2 % Windshield by 1.3 wt. Length of wind-9.9 15.1 23.7 19.65 shield

Design Data on Armor-piercing Projectiles

- 3 -

The 8" AP projectile has a greater percentage of cap and a harder cap than the other projectiles. The 14" AP projectile is only 4 calibers long, while the other projectiles are 4-1/2 calibers long. The 14" projectile has the windshield threads cut in the soft part of the skirt of the cap rather than in the hard upper part of the cap, which causes it to have a longer and heavier windshield than the other projectiles. Variations in the contours of the cap, the nose, and the cavity are apparent in Figure 1.

#### III. <u>HEAT TREATMENT</u>

According to the manufacturer's data the body of the 8" AP Mark 21-5 was hardened decrementally, after the conditioning treatment, by dipping nose downwards into lead at 1650°F and then water quenching; followed by an overall temper at 475°F. The triple alloy cap was hardened in the same way except that the overall temper was at 325°F. The 16" AP Mark 8-6 was given the same hardening treatment except that the cap was tempered at 450°F. The data available at the Proving Ground indicate that the 12" and 14" AP projectiles were heat treated similarly to the 16" AP projectile.

#### IV. <u>EXPERIMENTAL</u>

#### 1. <u>Chemical Analyses</u>

The results of chemical analysis of specimens from the caps and bodies of the projectiles are given in Table II. The values reported were obtained spectrochemically with the exception of those of carbon, phosphorus and sulfur which were determined by standard chemical procedures.

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### TABLE II

Chemical Compositions of Armor-piercing Projectiles Bodies

	<u>C</u>	<u>Mn</u>	P	<u>s</u>	<u>Si</u>	Ni	<u>Cr</u>	<u>Mo</u>	<u>Cu</u>
8" Mk. 21-	-5 .64	•38	.018	.018	•43	3.38	2.22	.08	.09
12" Mk. 18-	-1.59	•38	.023	.015	•37	3.20	2.16	•08	.10
14" Mk. 16-	.8 .58	.41	<b>°</b> 055	.015	•43	3.00	2.22	.06	.11
16" Mk. 8-	.60	<b>.</b> 34	.025	.017	<b>.</b> 26	3.33	2.30	.09	.10

#### <u>Caps</u>

	<u>C</u>	<u>Mn</u>	<u>P</u>	<u>s</u>	<u>S1</u>	<u>Ni</u>	Cr	<u>Mo</u>	<u>Cu</u>
8" Mk. 21-5	.81	.80	.021	<b>.</b> 022	.27	1.65	۰96	•33	.15
12" Mk. 18-1	.60	•39	.023	.014	.31	3.15	2.23	80،	.09
14" Mk. 16-8	.60	.46	.023	.020	。43	3.05	2.21	.06	.14
16" Mk. 8-6	.60	•35	.031	018ء	۰39	3.60	2.49	٥٥6	.10

The caps and bodies of all the projectiles were made of the same type of nickel chromium steel except the 8" AP projectile which had a high carbon triple alloy steel cap.

#### 2. Macroetch and Hardness Patterns

One projectile of each type was sent to the Naval Gun Factory for sectioning and surface grinding. The results of the macroetch and hardness survey of the ground halves are shown in Figures 2, 3, 4, 5 and 6.

The macroetch patterns of all projectiles were generally similar. The 12" AP body had the dark etching area extending all the way to the end of the base instead of ending above the bandscore, as for the other projectiles. The dark etching area at the base was about 7 points Rc greater in hardness than the light etching area. The 14" AP cap had a greater amount of light etching area than the

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caps of the other projectiles. The cracks visible in the hardened part of some of the projectiles are believed to have been caused by the splitting process.

All projectiles had similar hardness patterns except the 8" AP projectile which had a much harder cap than the others. Both Vickers and Brinell hardness tests were made on the same 8" AP projectile so that the values could be compared. As would be expected the Vickers values were somewhat higher than the Brinell numbers in the higher ranges but otherwise the hardness patterns obtained by either method were similar.

#### 3. Impact Properties of the Base

Specimens for impact tests were taken from the base of the unground half of each projectile. The results of these tests are shown in Table III.

#### TABLE III

	<u>8" AP</u>	<u>12" AP</u>	14" AP	<u>16" AP</u>
Temp. °C	Ft. lbs.	Ft. 1bs.	Ft. 1bs.	Ft, 1bs.
+100	55 F	45 F	69 F	65 F
+ 50	57 F	48 F	57 F	58 F
0	45 F	<b>4</b> 8 F	47 G10%	56 F
- 50	26 <b>G</b> 50 <b>%</b>	24 G80%	21 G100 <b>%</b>	39 <b>G40%</b>
- 78	17 G100%	20 G100%	17 G100%	20 G100 <b>%</b>

Results of Charpy V-notch Impact Tests

F = All fibrous fracture. G = Grainy fracture, percent.

All the projectiles have about the same impact properties. The hardness of the impact specimens varied between 23 and 27 Rc. Impact specimens were taken from the light etching part of the base of the 12" AP projectile.

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#### 4. <u>Microstructure</u>

Photomicrographs were made of samples taken from the nose and skirt of the caps and from the base and nose of the bodies of all the projectiles. These are shown in Figures 7, 8, 9 and 10. The microstructure in the nose of the caps and in the nose of the bodies consisted of slightly tempered martensite while the microstructure of the skirt of the caps and base of the bodies consisted of spheroidized carbides in a ferritic matrix.

### V. <u>DISCUSSION</u>

## 1. <u>Cap Hardness</u>

The present standard 12", 14" and 16" AP projectiles have caps of relatively low hardness 550 Brinell (carbide ball). As was pointed out in the introduction the penetrative ability of the 8" AP was increased by increasing the cap hardness, so that increasing the cap hardness of the 12", 14" and 16" AP projectiles might increase their penetrative ability.

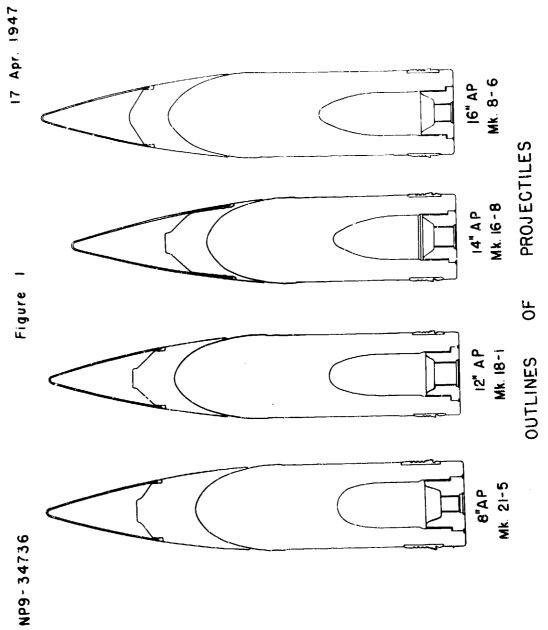
## 2. <u>Body Hardness</u>

Higher nose and body hardness for the subject projectiles might be desirable. With many steels the breaking strength on impact is greater when the steel is tempered at 350°F than when the steel is tempered at 450°F even though the steel is harder when tempered at 350°F. Therefore, if the bodies of the subject projectiles were given a final tempering at 350°F instead of 450-475°F as at present, the ballistic performance might be improved. Tempering at 350°F would give harder steel that is likely to have greater impact strength.

#### 3. <u>Composition</u>

Most Navy projectiles regardless of size are made of steel of the same analysis (.60C, 2.2 Cr and 3.2 Ni). Considering hardenability only, it may be that a more highly alloyed steel would produce better properties in the larger projectiles (12", 14" and 16").

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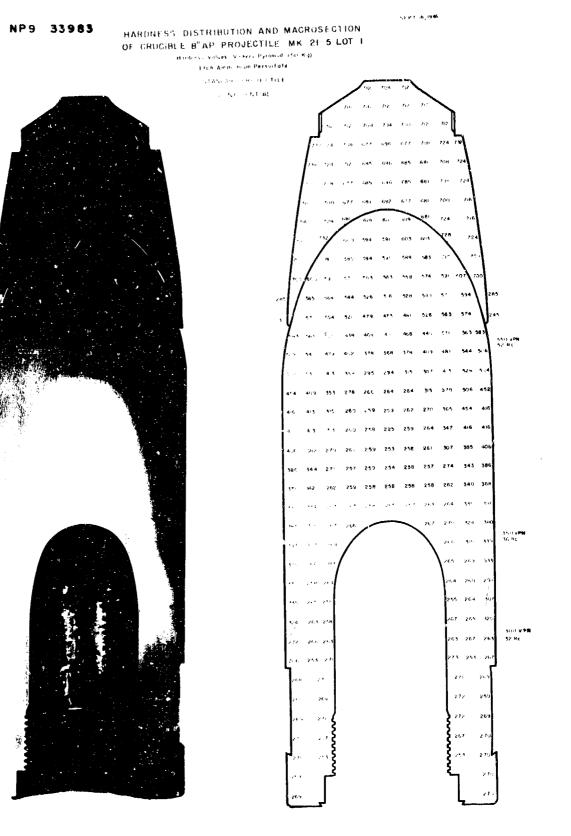
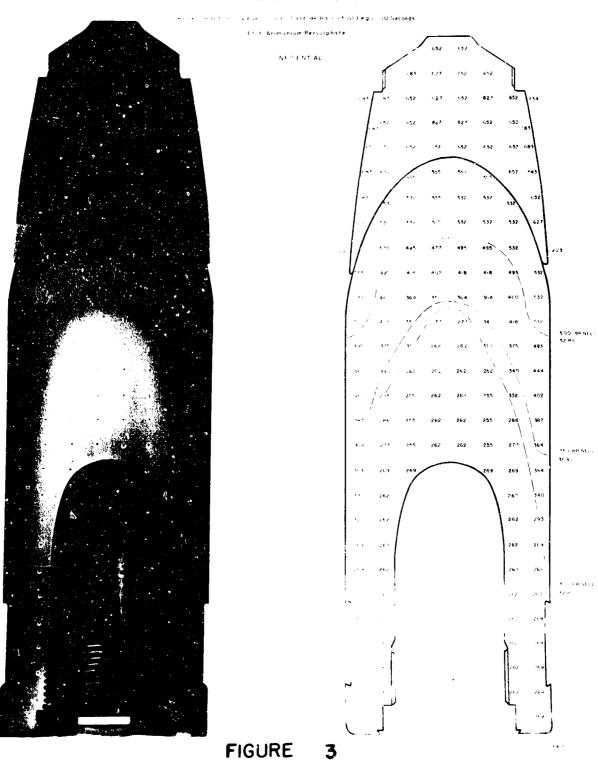


FIGURE 2

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# HARENESS ESTRIBUTION AND MACROSECTION

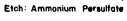
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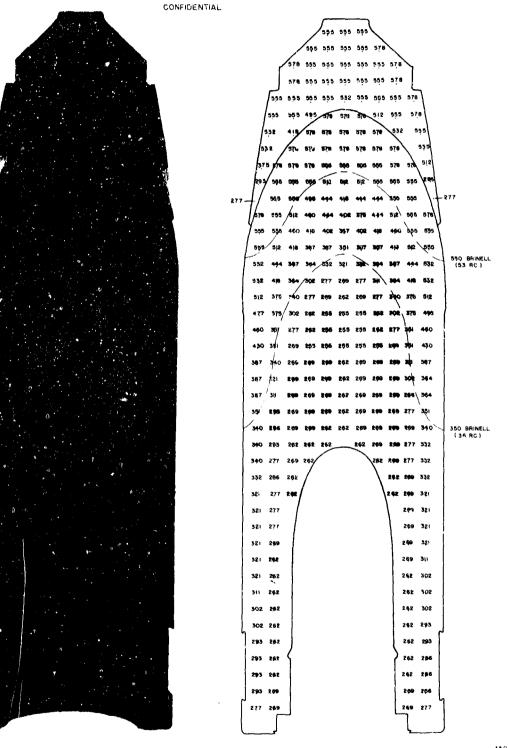


NP9 - 34004

Nov. 13, 1946

### HARDNESS DISTRIBUTION AND MACROSECTION OF CRUCIBLE 12" A.P. PROJECTILE MK. 18-1 LOT 21 Brinell Hardiness Values: 10 mm. Carbide Ball - (3000 kg)-10 Seconds





FIGURE

4

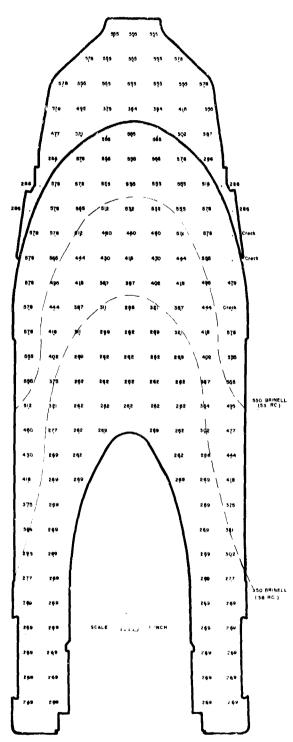
3507

NP9 - 34011

# HARDNESS DISTRIBUTION AND MACROSECTION OF CRUCIBLE 14" AP PROJECTILE MK. 16-8

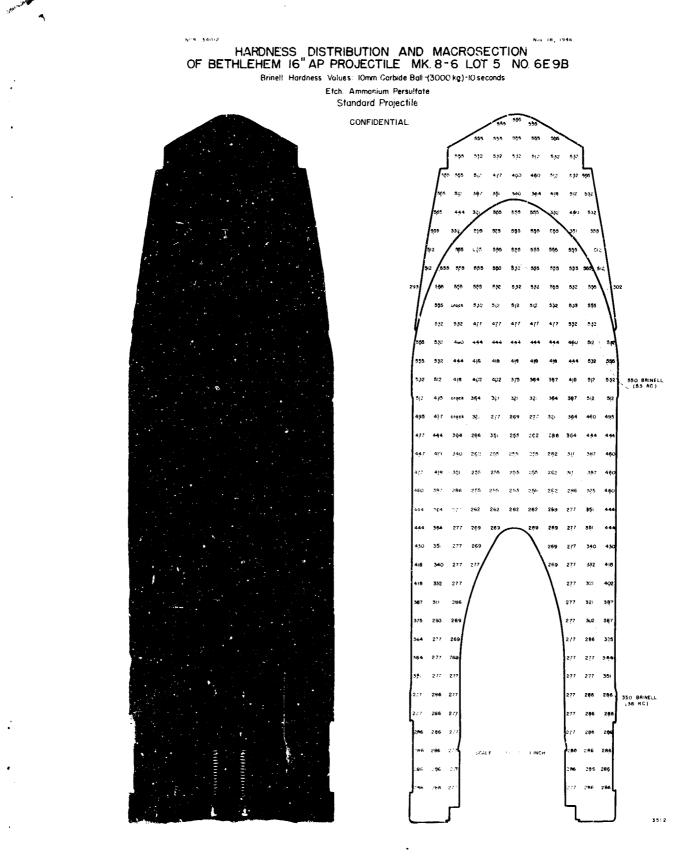
Brinell Hardness Values: IOmm. Carbide Ball -(3000kg.)- IO seconds Etch: Ammonium Persulfate Standard Projectile





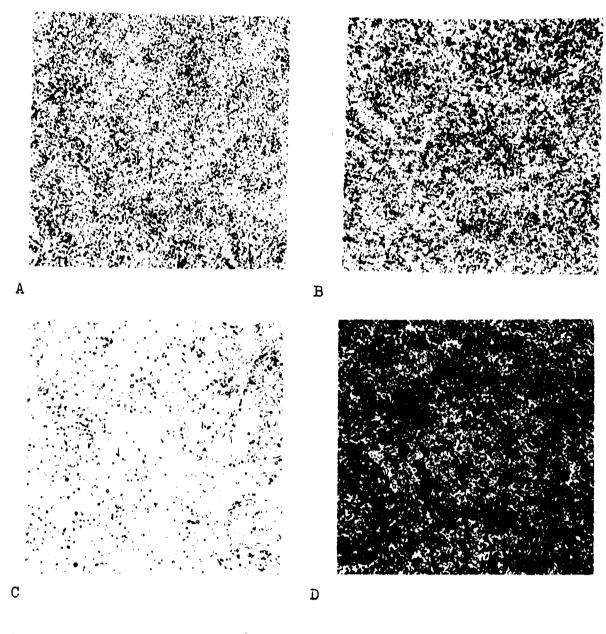
FIGURE

5



FIGURE

6

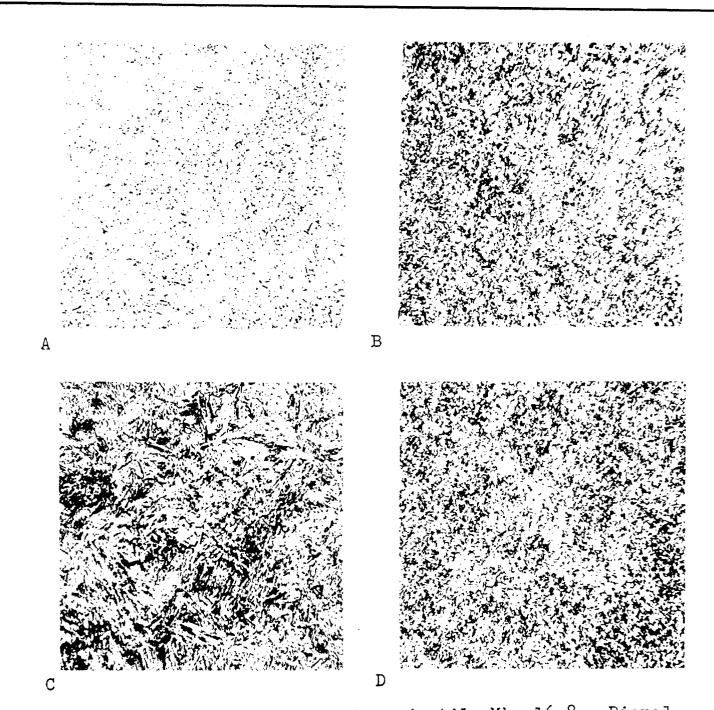


Photomicrographs of the 8" AP projectile Mk. 21-5. Picral Etch, 750X

# <u>Hardness</u>

B. C.	NP9 34196 NP9 34211	<ul> <li>Nose of body, tempered martensite</li> <li>Base of body, spheroidized structure</li> <li>Nose of cap, tempered martensite</li> <li>Skirt of cap, spheroidized structure</li> </ul>	555 BHN 269 BHN 652 BHN 269 BHN
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Figure 7



Photomicrographs of the 14" AP projectile Mk. 16-8. Picral Etch, 750X

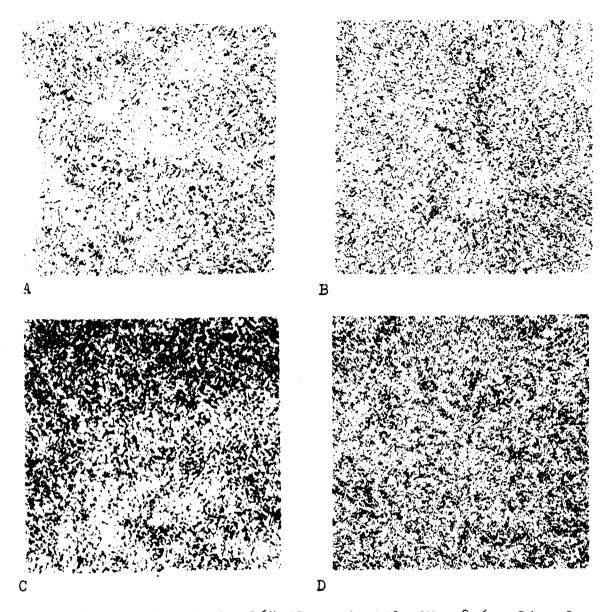
# <u>Hardness</u>

Α.	NP9 34435	- Nose	of body, tempered	martensite	555	
B.	NP9 34214	- Base	of body, spheroid:	ized structure	269	BHN
C	NPG 34200	- Nose	of cap, tempered n	nartensite	555	
D.	NP9 34215	- Skir	; of cap, spheroid:	ized structure	286	BHN

Figure 9

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Photomicrographs of the 16" AP projectile Mk. 8-6. Picral Etch 750X

# <u>Hardness</u>

Α.	NP9	34437	-	Nose	of	body,	tempered	mart	tensite	555	BHN
B.	NP9	34219		Base	of	body,	spheroid	ized	structure	286	BHN
							tempered			555	BHN
										293	BHN

Figure 10